



Industrial Technology
Research Institute

新材料微晶矽鍺薄膜太陽電池

綠能與環境研究所 薄膜太陽電池研究室

張佳文 研究員

changcw@itri.org.tw

03-591-5371

2010/11/15 PM3:00~5:40

台大博理館 101演講廳

Outline

- Introduction
- Conventional thin film silicon solar cell
 - Challenges in a-Si:H and a-Si:H/ μ c-Si solar cell
- A new concept of thin film μ c-Si_{1-x}Ge_x:H solar cell
 - Advantages and opportunities
 - Challenges
- Properties of μ c-Si_{1-x}Ge_x:H thin film
 - ESR spin densities (neutral dangling bonds)
 - Carrier densities
- Conclusions



Si thin film

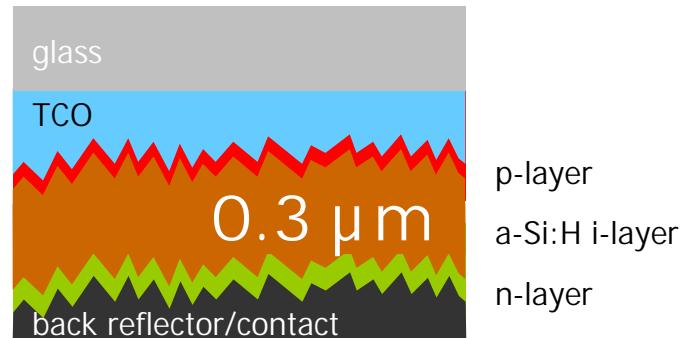


Si bulk-type



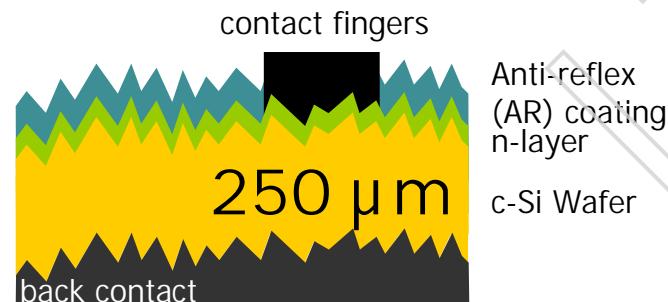
Why Silicon Thin Film for Solar Cells?

Thin
Film



1000 x thicker

Wafer



Silicon Thin Film Technology:

- low material consumption 0.3 μm to 3 μm
- established large area
- deposition techniques (Flat Panel Industry)
- low process temperatures (< 300°C)
- low-cost substrates (glass, plastics, metal, stainless)
- higher energy yield
- low energy pay back time

Silicon-based as raw material:

- abundantly available
- non-toxic & ecologically harmless



Advantages of Si thin film solar cell

Cost Performance → Grid Parity

Building Integration

Aesthetic Design, Wide Applications and See-Through Capability

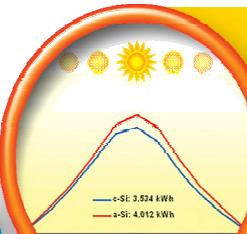


Sustainable Product

Environmental Friendly



Shorter Energy Payback Period



Higher Yield (kWh/kWp)

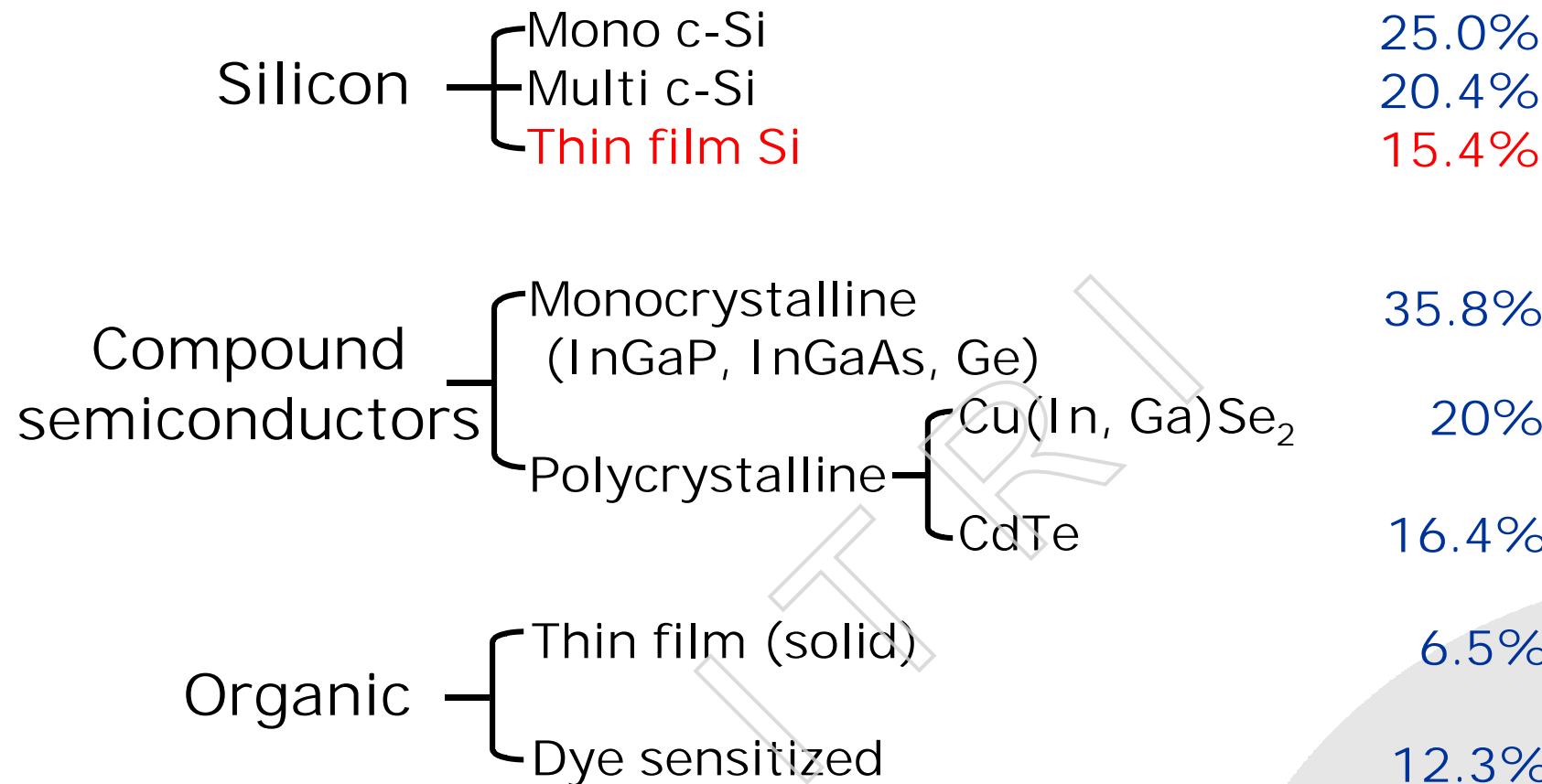
Better Return on Investment

Less Si Materials Used

Scale Up Capability

Monolithic

PV technologies



Quantum effect—Quantum dot, multi-quantum-well

Target

日本NEDO計畫電池及模組之性能目標(轉換效率%)

個別技術的開發	太陽電池 ¹⁾	現狀		2017年		2025年		2050年
		模組 (%)	電池 ⁵⁾ (%)	模組 (%)	電池 ⁵⁾ (%)	模組 (%)	電池 ⁵⁾ (%)	模組 (%)
40%超 高效率太 陽電池 (追加 開發)	結晶矽 ²⁾	~16	25	20	25	25	(30)	40%超 高效率太 陽電池 (追加 開發)
	矽薄膜	~11	15	14	18	18	20	
	CIS系	~11	20	18	25	25	25	
	化合物系 ³⁾	~25	41	35	45	40	40	
	染料敏化	-	11	10	15	15	15	
	有機系 ⁴⁾		5	10	12	15	15	

1) 電池的技術指標以實驗室小面積規格為主。模組則為實用化技術階段。

工研院 張佳文翻譯 (2010)

2) 結晶矽不予以單晶、多晶分類，以使用矽基板之太陽電池為設定。

3) 集光時的轉換效率。

4) 新型太陽電池之有機系太陽電池也為設定之開發目標。

5) 為達成模組目標所設定之電池最低轉換效率。

Thin Film competition

- **Si(Ge)**

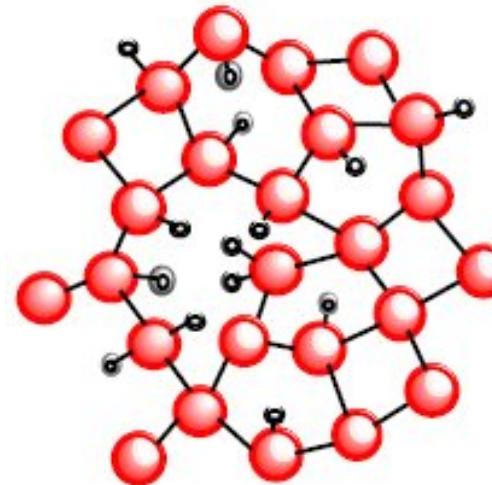
Abundance, environment safety, ubiquitous
Limited efficiency, equipment cost (throughput)
Multi-junction

- **CdTe**

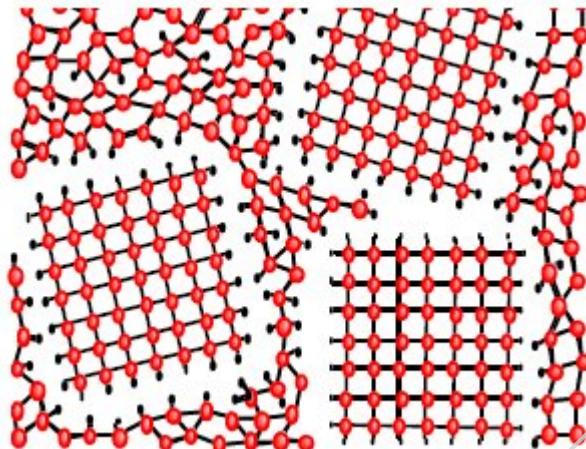
Low cost, process simply, high throughput
Toxicity (Cd), rare element (Te), recycling problem
Single junction

- **CIGS**

High potential efficiency (~20%), tunable Eg
Toxicity (In compound), rare element (In)
Single junction

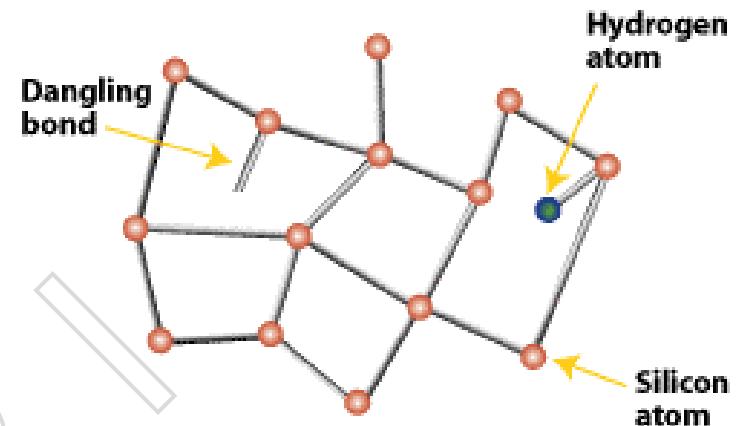


a-Si:H



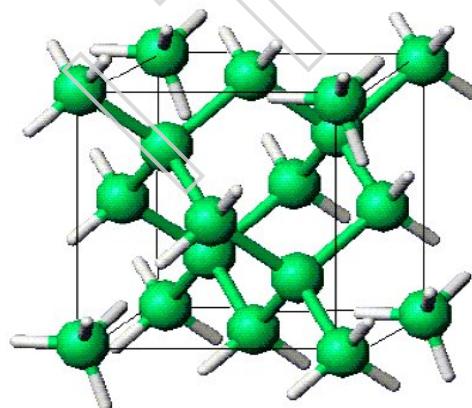
μc-Si:H

Schematic structure



Dangling bond

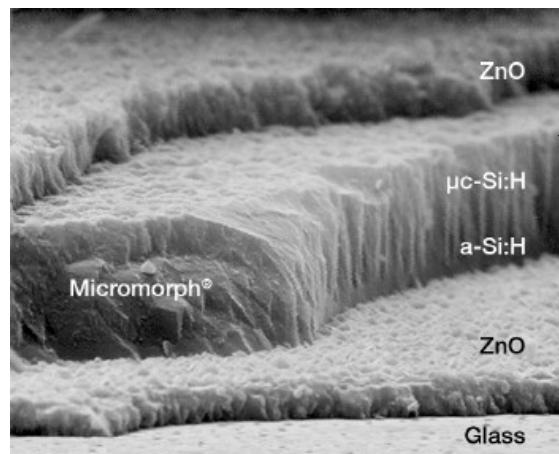
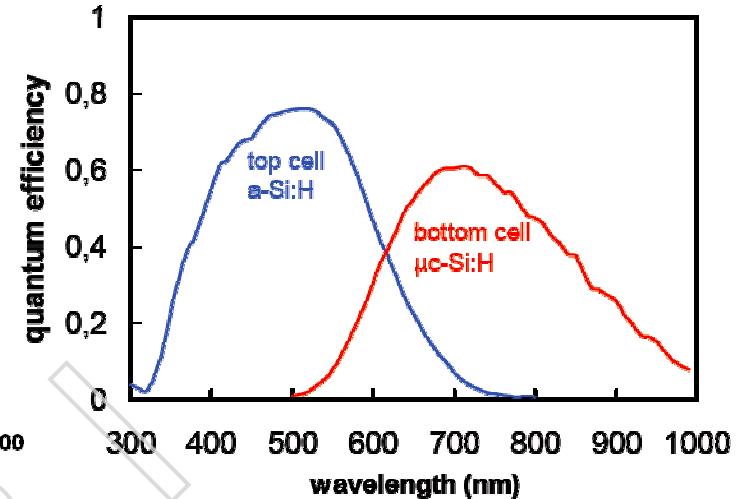
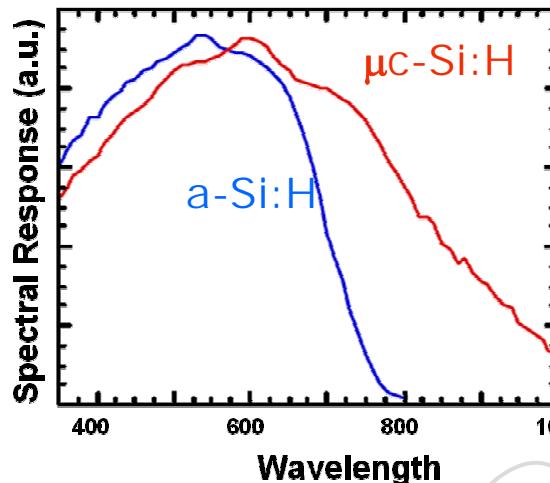
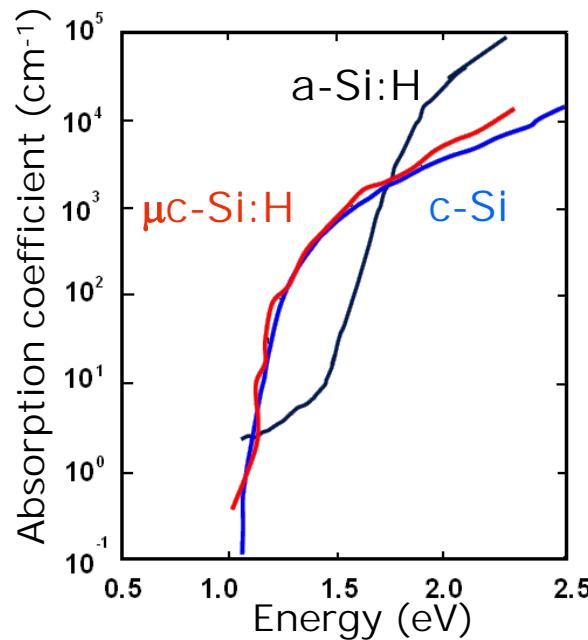
Si, Ge
Diamond structure





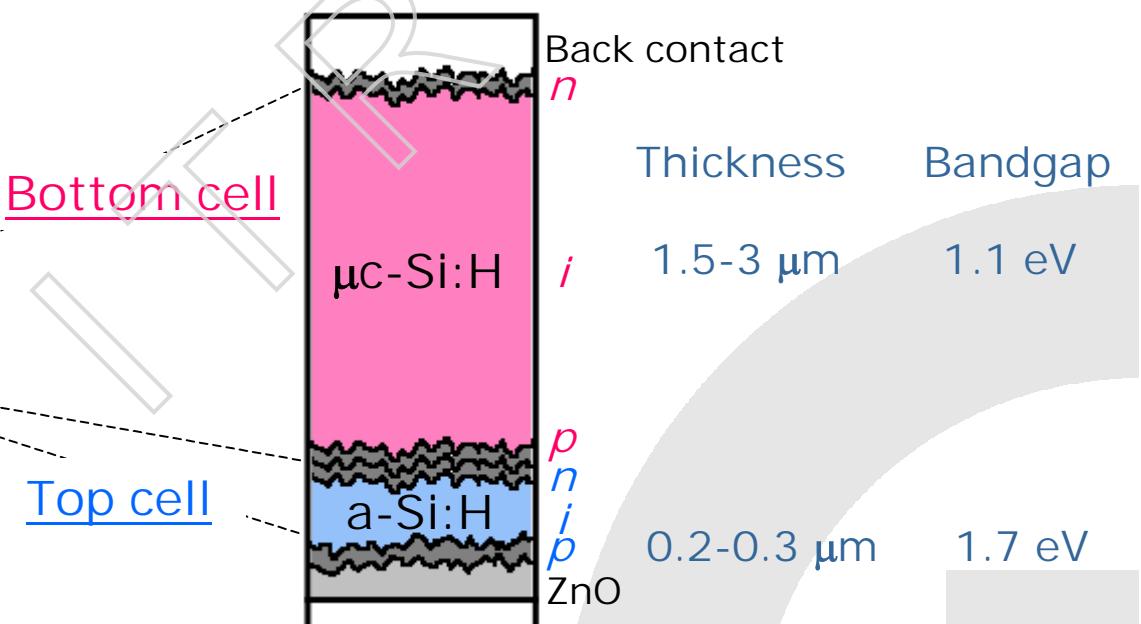
Micromorph a-Si:H/ μ c-Si:H tandem solar cell concept

(IMT Neuchâtel 1994)



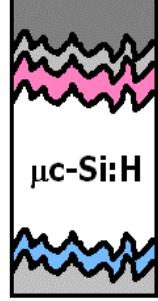
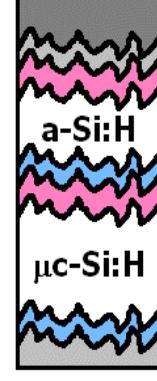
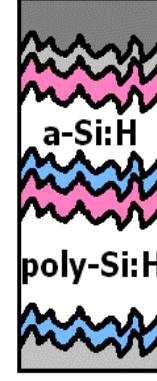
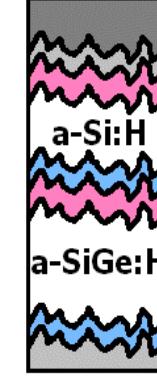
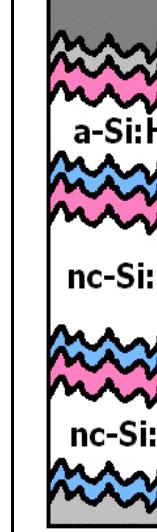
Courtesy of IMT

SEM Micromorph



Benchmarks

Small area cell (<1 cm²)

single		tandem			triple		
							
a-Si	μc-Si	a-Si/ μc-Si	a-Si/ poly-Si	a-Si/ a-SiGe	a-Si/ a-SiGe/ a-SiGe	a-Si/ nc-Si/ nc-Si	a-Si/ a-SiGe/ nc-Si
IMT Oerlikon	Kaneka	IMT Oerlikon	Kaneka	BP Solar, Sanyo	Uni-Solar	Uni-Solar	Uni-Solar
10.1% (s) pin	10.9% (i) pin	13.7% (i) pin 9.8 (i) nip	14.8% (i) 12% (s) pin	11.6% (i) 10.6% (s) pin	15.2% (i) 13.0% (s) nip	13.8% (i) 13.2% (s) nip	14.5% (i) 12.6% (s) nip



Materials cost

gas price (per 1 liter)

$SiH_4 : GeH_4 : H_2 = 1: 100: 0.05$

Ref. A. Baumann, 2004

Process	Material	Utilization rate (%)	Cost (\$/kg)	Thickness (μm)	Cost (\$/W)
CdTe sublimation (commercial)	CdTe	75	170	4	0.05
CdTe electrodeposition (pilot line)	Te	95	250	2	0.02
In-line a-Si GD (commercial)	Ge	10	3000	1	0.12
Box carrier (batch) a-Si (commercial)	Ge	25	3000	1	0.05
High-rate a-Si (experimental)	Ge	10	3000	1	0.12
High-rate CIGS evaporation (experimental)	In	50	400	2	0.03
Sputtering CIGS (experimental)	In target	75	800	2	0.043
Silicon film™(experimental)	Si	75	20	50	0.03
Single crystal silicon	Si (feedstock)	45	20	320	0.32

Ref. B.A. A'ndersson, Energy 23 (1998) 407-411.

Table 1. Materials requirements and indicators for the solar cells in four solar energy systems, each based on a specific thin-film technology supplying 100,000 TWh/yr.

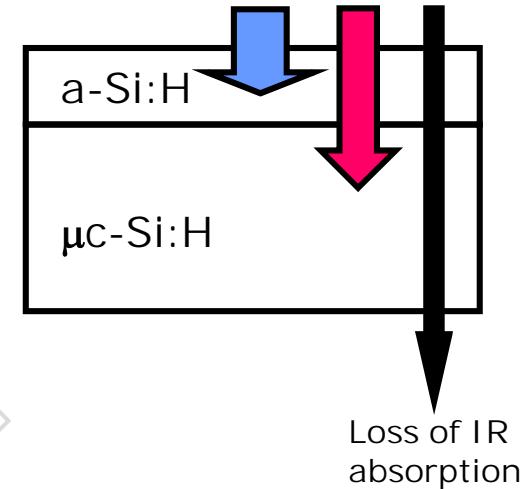
Materials requirements (g/m^2)	Total material requirements ^b (Gg)	Total material requirements /reserves ^c	Total material requirements /max. resources ^d	Annual material requirements ^e /refined materials ^f	Potential losses ^g / weathered amounts ^h	Material cost share ⁱ (%)
<i>α-SiGe^a</i>						
Sn	3.3	1700	0.20	0.004	0.079	0.04
Ge	0.22	110	51	0.0003	21	0.5
Si	0.54	270	Negligible	Negligible	0.0031	0.002
Al	2.7	1400	0.00032	Negligible	0.00075	0.008

A new concept of thin film μ c-Si_{1-x}Ge_x:H solar cell

- Advantages and opportunities
- Challenges

Challenges of Si thin film solar cells

- Single junction a-Si:H
- Tandem junction a-Si:H/ μ c-Si:H
 - ➔ only absorbs VIS-near IR light
 - ➔ lower absorption coefficient in long wavelength
 - ➔ Light absorption is not enough in the infrared



New concept of multi-junction solar cell

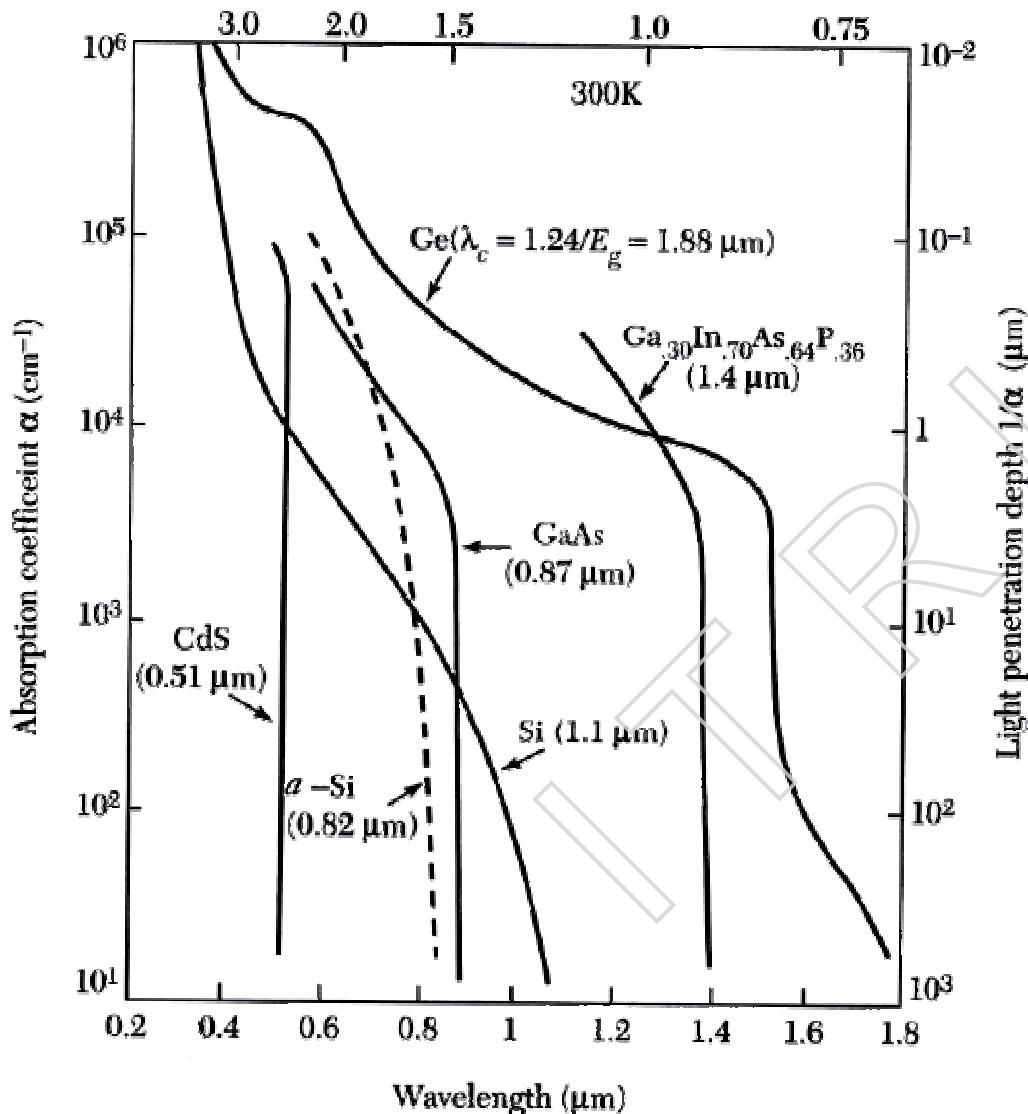
New candidate for a bottom cell material

μ c-Si_{1-x}Ge_x:H

Ref. G. Ganguly, M. Kondo, A. Matsuda, APL, 69 (1996) 4224.

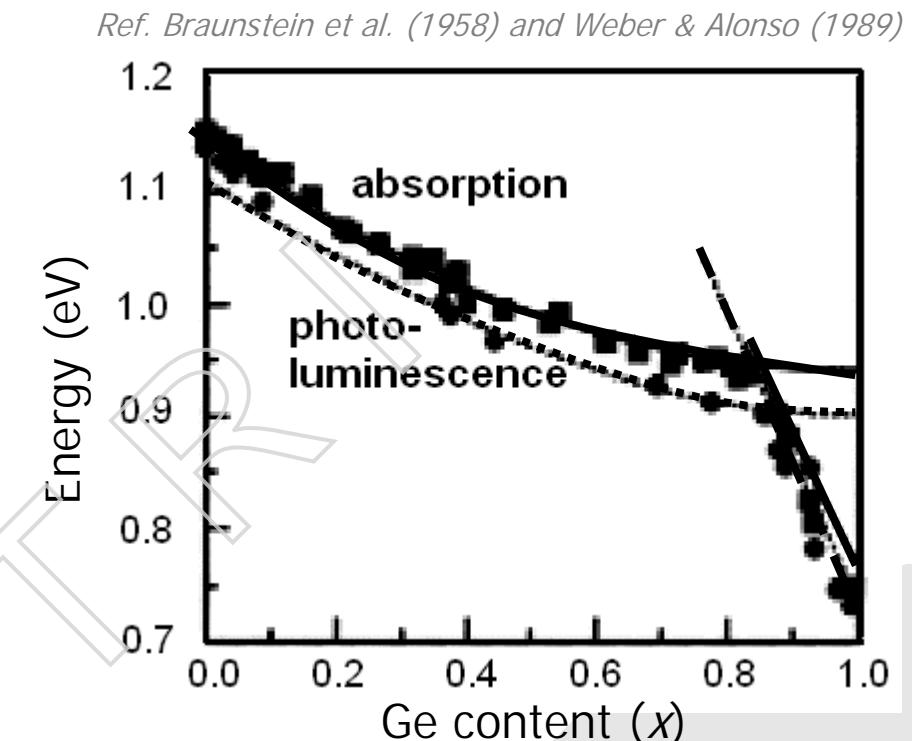
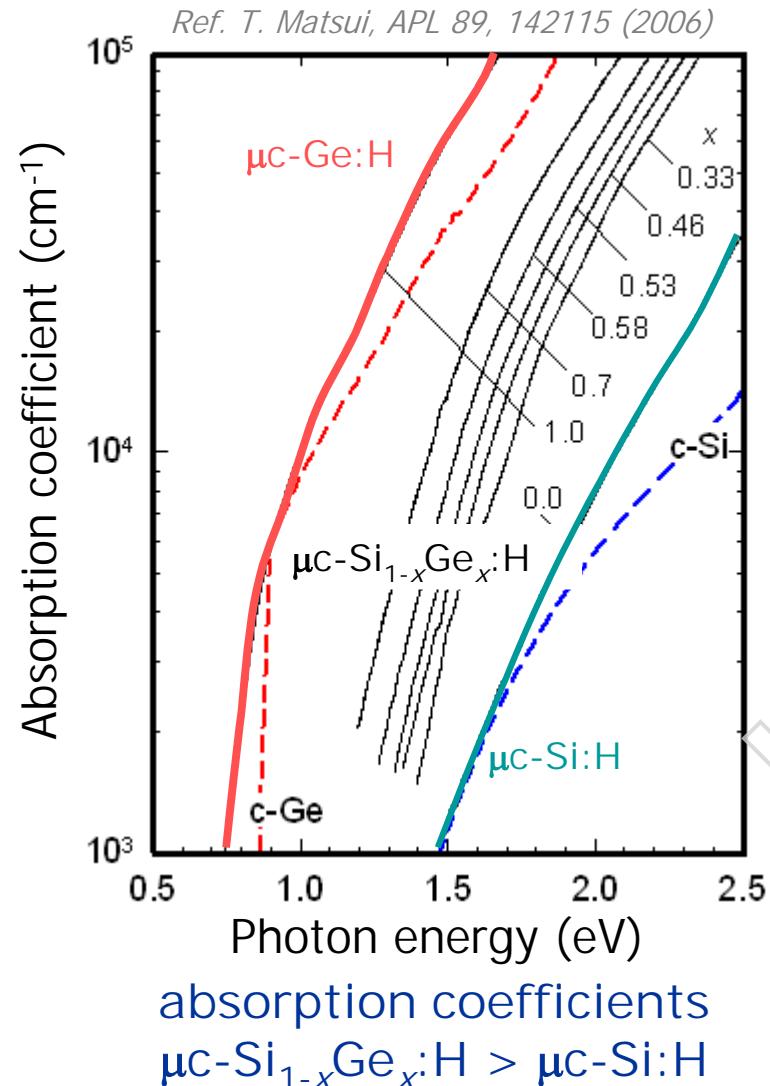


Absorption coefficient



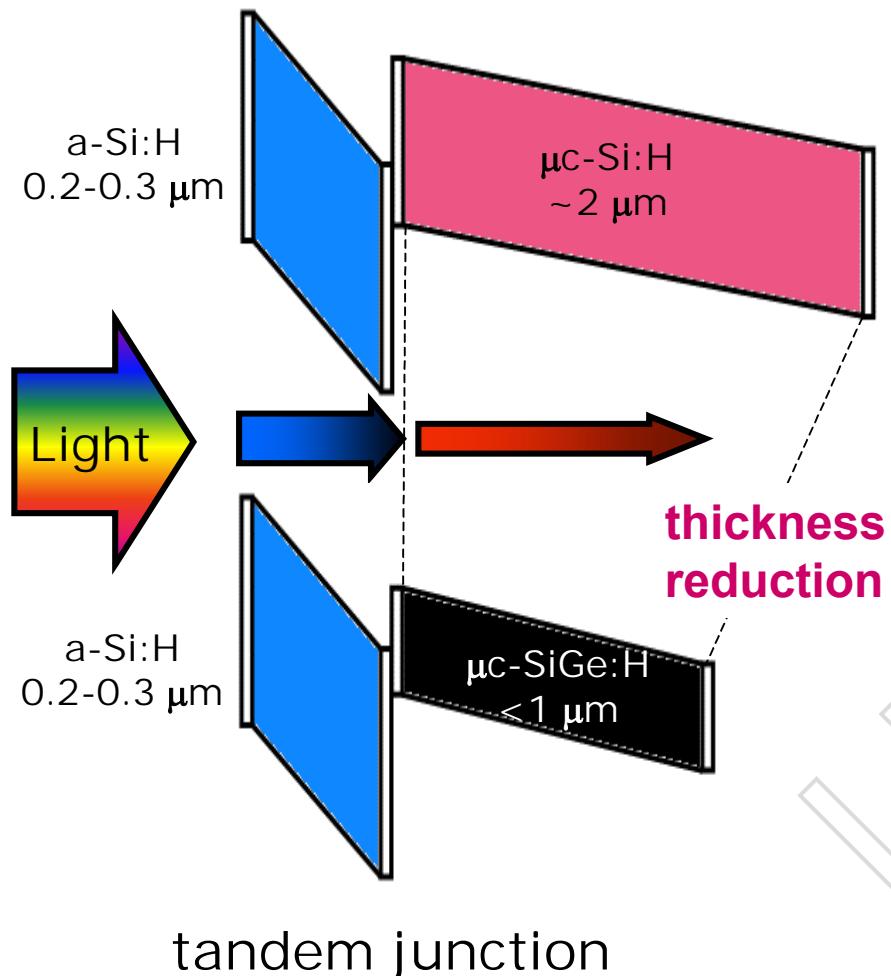
$$A = 1 - \exp(-\alpha d)$$
$$\alpha_{\text{Ge}} > \alpha_{\text{Si}}$$

$\mu\text{c-Si}_{1-x}\text{Ge}_x:\text{H}$ as narrow gap absorber for multi-junction solar cell



Tunable bandgap of
 $\mu\text{c-Si}_{1-x}\text{Ge}_x:\text{H}$
 $E_g \sim 1.1 - 0.7\text{ eV}$

Why Silicon-Germanium Thin Film?



$$\mu_{\text{ini}} = 11.2\% \text{ @ } d_{\text{bottom}} = 0.9 \text{ } \mu\text{m}$$

Ref. T. Matsui and M. Kondo, AIST (2009)

Thickness $\mu\text{c-Si:H} >> \text{a-Si:H}$
 $\mu\text{c-Si:H}$ is 10 times thicker than a-Si:H

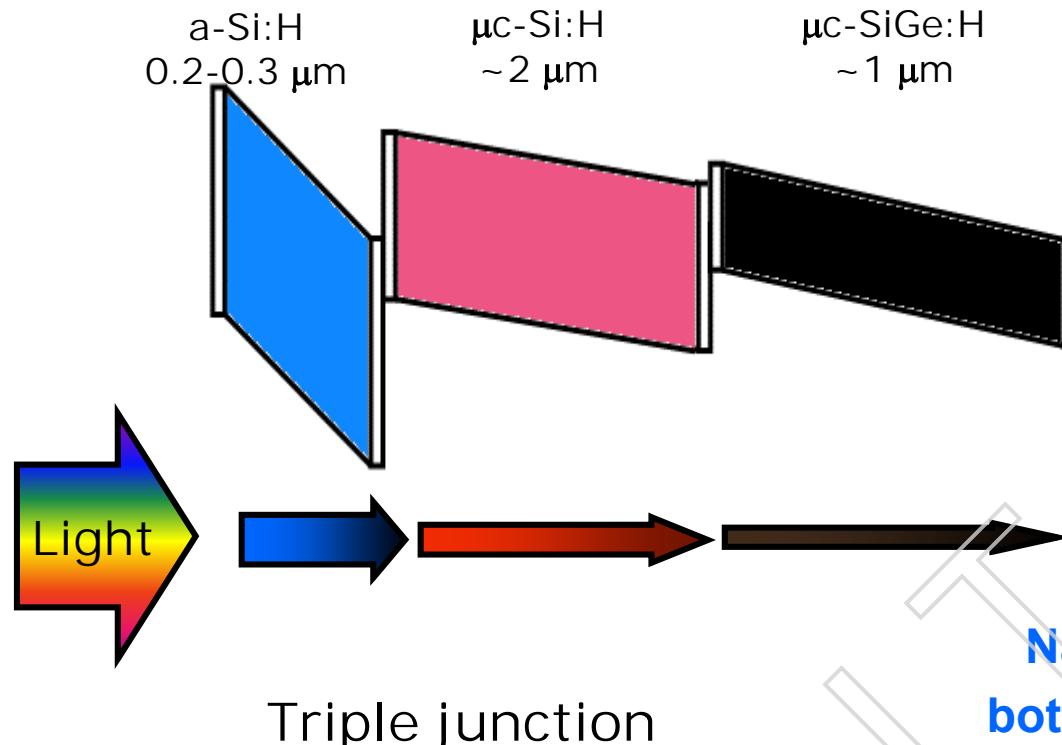
Deposition rate $\mu\text{c-Si:H} \ll \text{a-Si:H}$
 High-rate deposition technique
 VHF and HPD (high-pressure depletion) plasma CVD

Bottom cell should be made as thin as possible



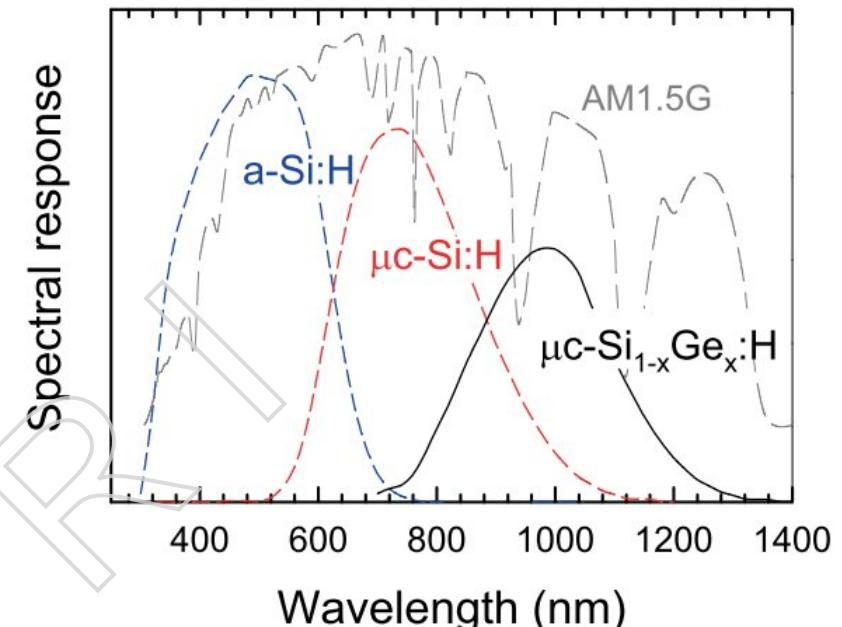
- Short deposition time
 \Rightarrow Industrially advantageous
- Less film thickness
 \Rightarrow Acceptable to flexible solar cells
- Increased long wavelength and IR absorption
 \Rightarrow Enhanced solar cell efficiency

Advantages of μ c-SiGe thin film solar cell



$$\mu_{ini} = 11.6\%$$

Ref. T. Matsui and M. Kondo (2010/08 AIST成果發表會)



Narrow bandgap material μ c-SiGe:H
bottom cell application in triple junction

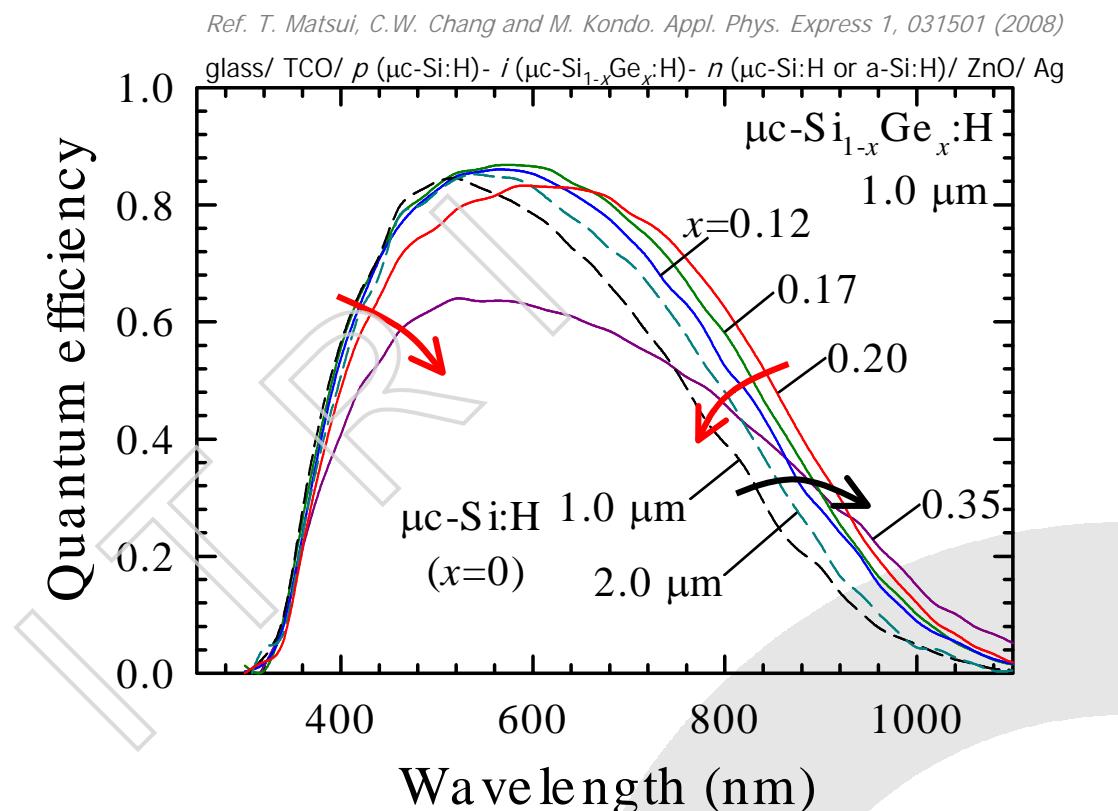
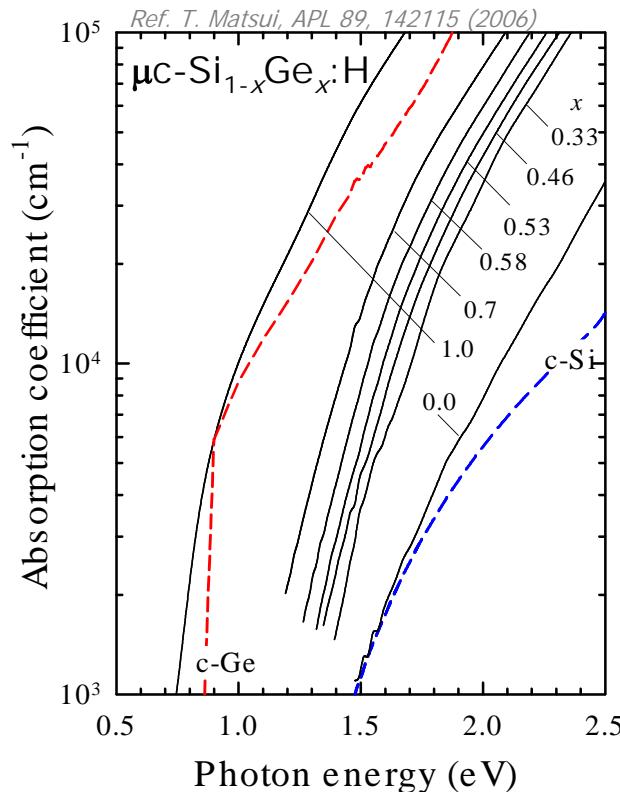
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$a\text{-Si:H}/\mu\text{c-Si:H}/\mu\text{c-Si}_{1-x}\text{Ge}_x\text{:H}$

Increased NIR absorption
Enhanced solar cell efficiency

Issues of $\mu\text{c-Si}_{1-x}\text{Ge}_x:\text{H}$

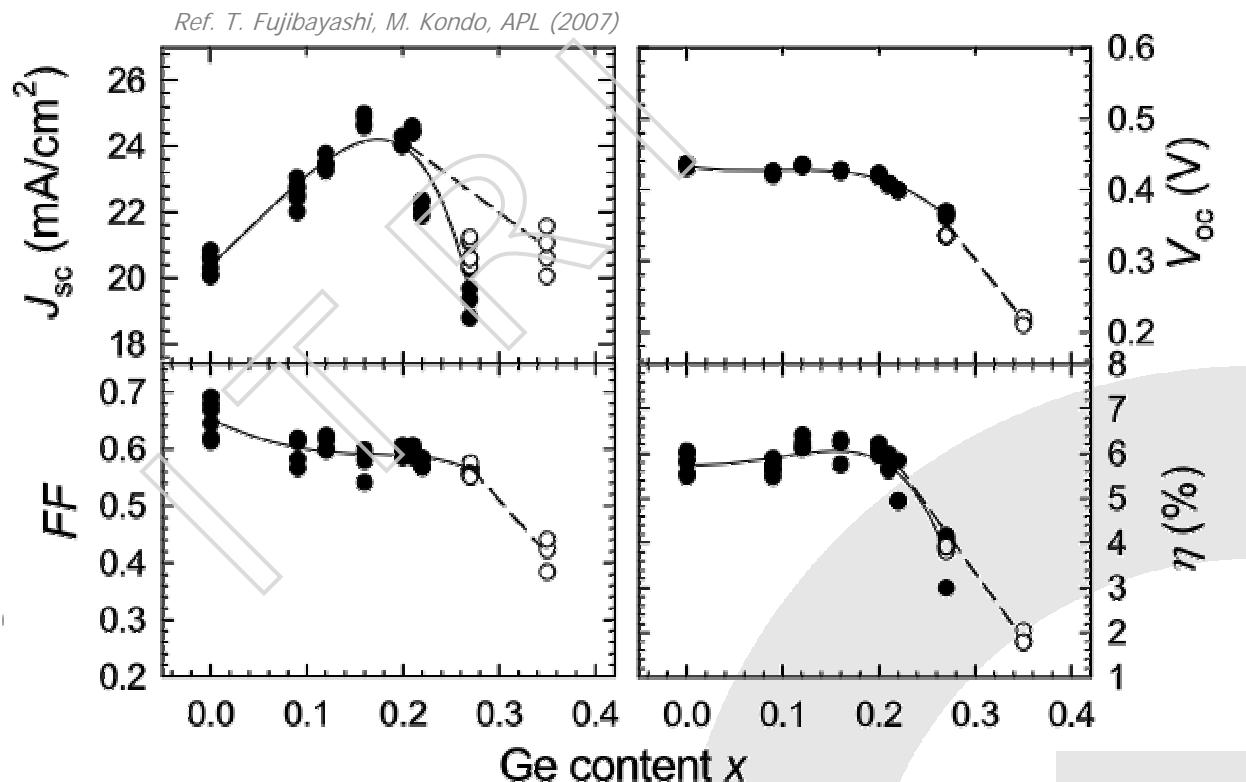
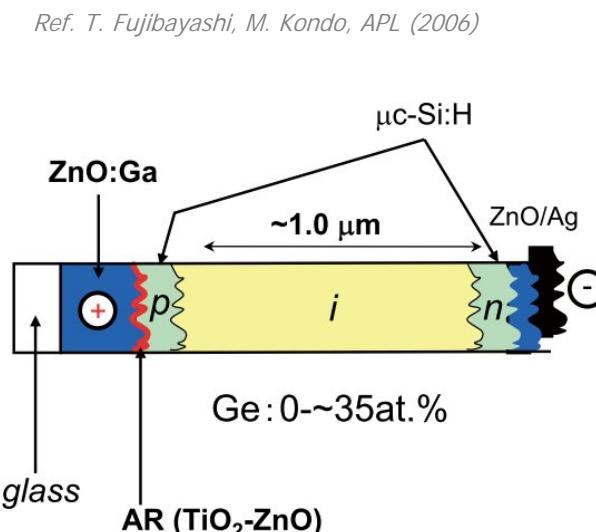
- $x = 0\text{-}0.2 \Rightarrow$ IR sensitivities increase monotonically
- $x > 0.1\text{-}0.2 \Rightarrow$ IR sensitivities of $\mu\text{c-Si}_{1-x}\text{Ge}_x:\text{H}$ ($1\mu\text{m}$) $>$ $\mu\text{c-Si:H}$ ($2\mu\text{m}$)



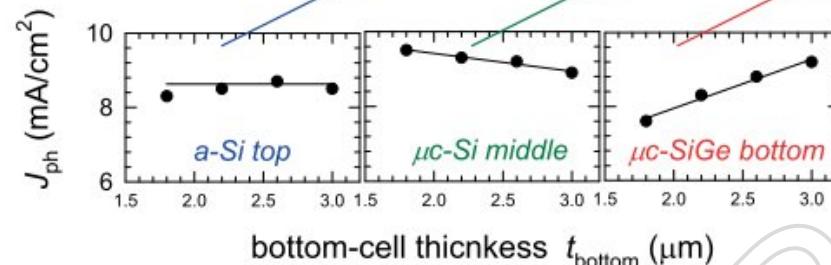
$x > 0.2 \rightarrow$ solar cell performance degrades drastically

Problems of $\mu\text{c-Si}_{1-x}\text{Ge}_x:\text{H}$

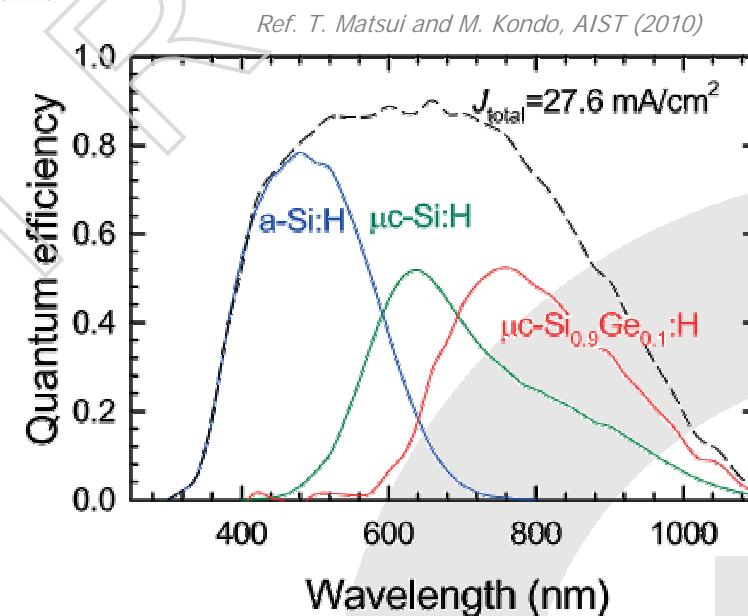
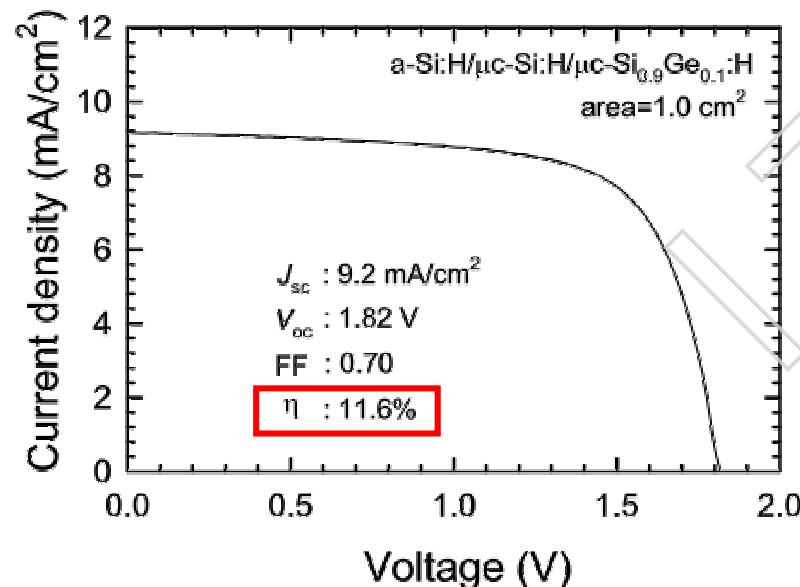
- $x = 0.15\text{-}0.2 \Rightarrow J_{sc} \text{ max.}$
 $\Rightarrow J_{sc} \text{ gain } \sim 5 \text{ mA/cm}^2$
- $x > 0.2 \Rightarrow \text{solar cell performances decrease}$
(charged Ge dangling bonds increase ?)



Performance of triple junction of a-Si:H/ μ c-Si:H/ μ c-Si_{1-x}Ge_x:H



Ref. T. Matsui and M. Kondo, AIST (2010)



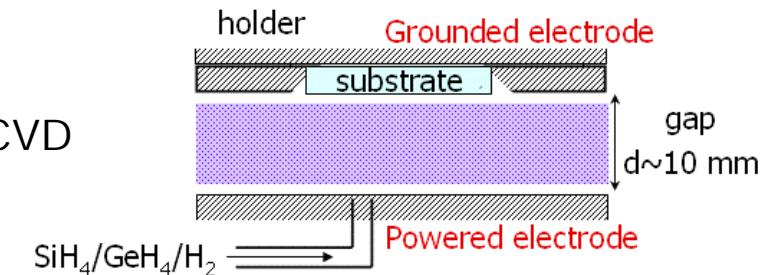
Ref. T. Matsui and M. Kondo, AIST (2010)

Properties of μ c-Si_{1-x}Ge_x:H thin films

- ESR spin densities
(neutral dangling bonds)
- Carrier densities

Methods and Experiments

VHF PECVD



- Raman measurement -

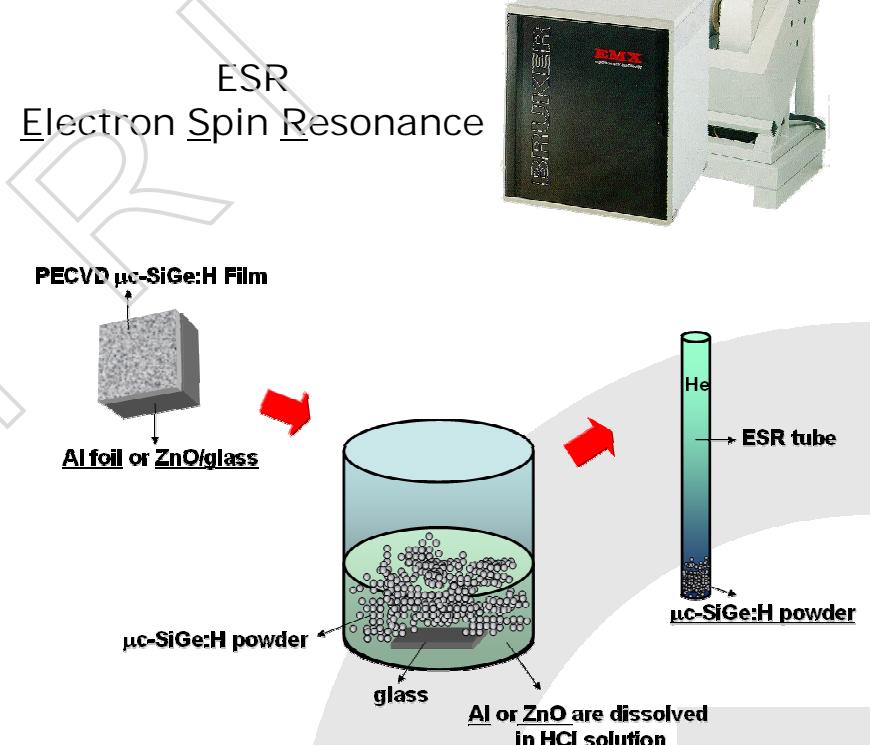
- He-Ne laser ($\lambda = 633$ nm)

- ESR measurement -

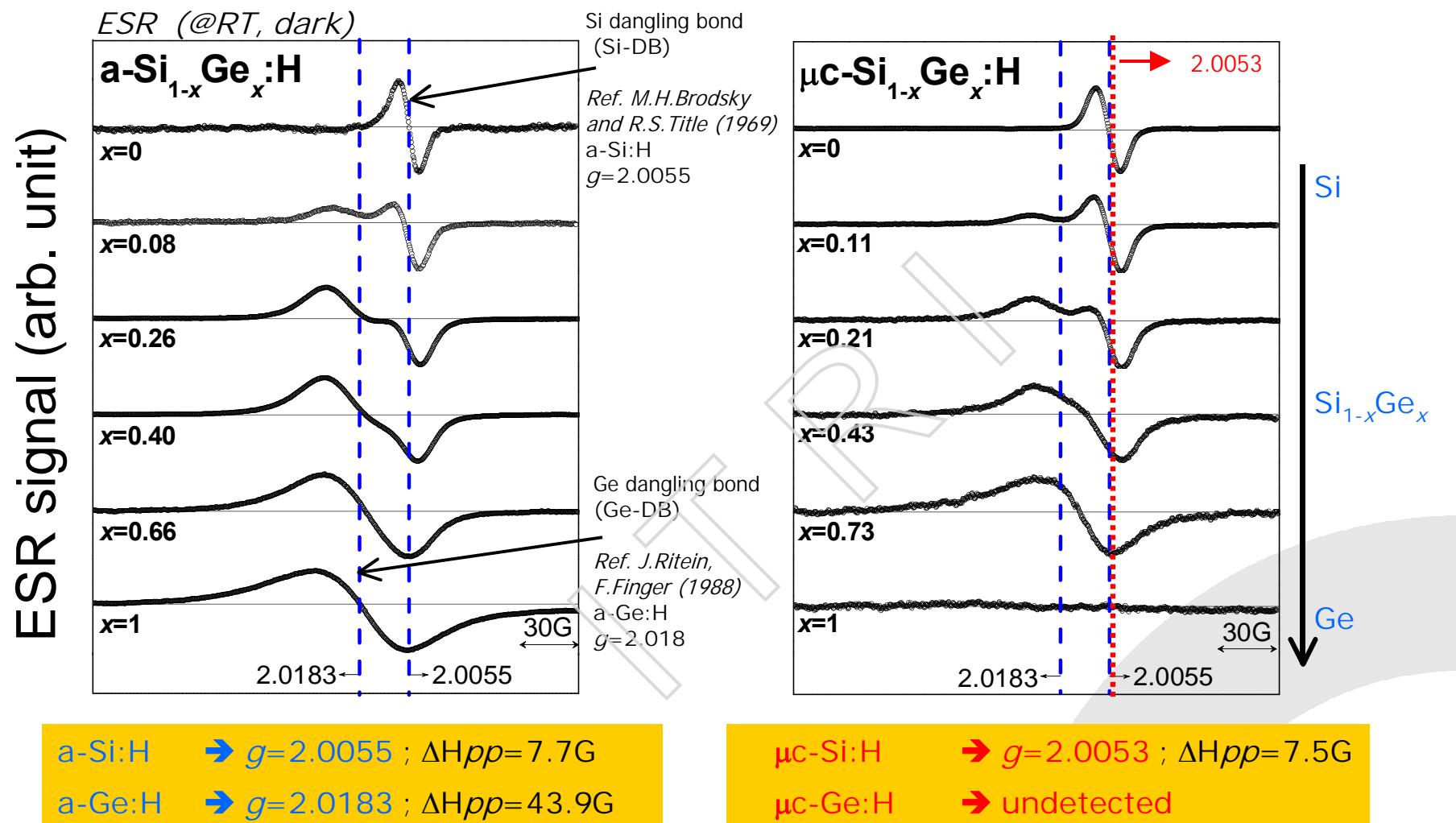
- Powdered samples (10-40 mg) were obtained by HCl etching of μ c-SiGe:H films grown on ZnO/glass substrates.
- ESR measurements were performed at room temperature in an X-band (~9.5 GHz) spectrometer.
- Defect density was determined by measuring the spin numbers of the $\text{CuSO}_4 \cdot 5\text{H}_2\text{O}$ used as a standard sample.

- Hall-effect measurement -

- Hall-effect measurements were performed at room temperature using Van der Pauw method.

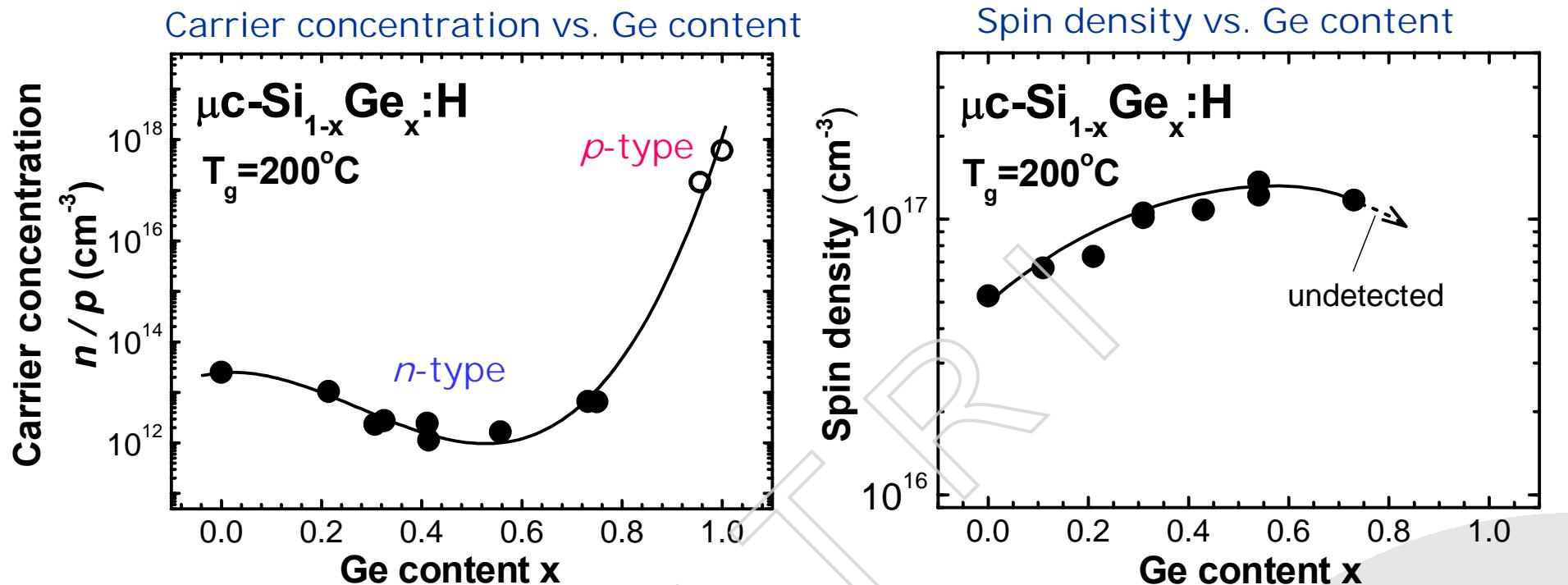


ESR spectra of a-SiGe:H and μ c-SiGe:H thin films for various Ge contents



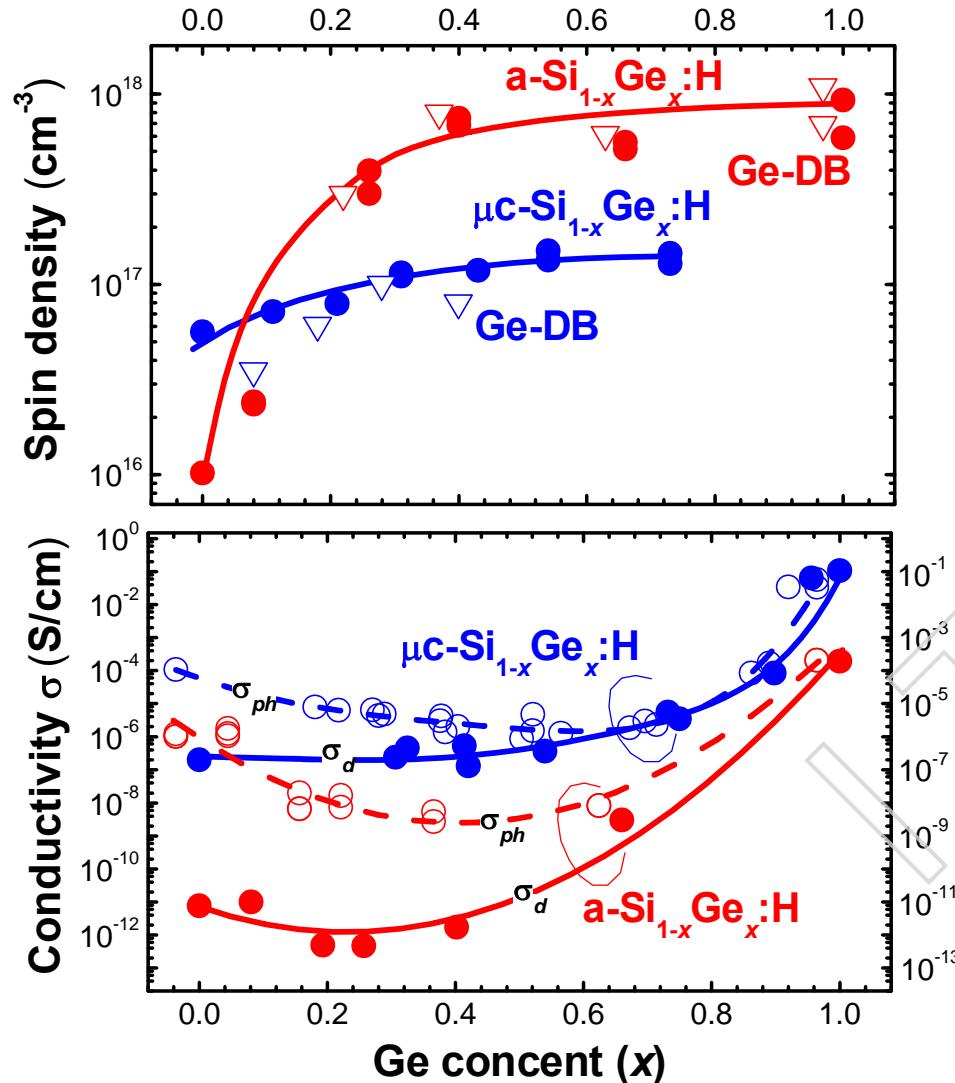
C.W. Chang, Ph.D thesis, Tokyo Institute of Technology, 2009.
C.W. Chang*, T. Matsui, M. Kondo, J. Non-cryst. solids, 354 (2008) 2365.

Carrier concentration and neutral defect density of μ c-SiGe



- Ge-rich → strong *p*-type: Ge incorporation induces an acceptor state generation (probably at grain boundary)
 - ⇒ Fermi level shift toward the valence band edge.
- ESR signal was undetected when the carrier concentration became comparable to dangling bond density.

Electrical properties of a-SiGe and μ c-Si_{1-x}Ge_x:H with various Ge contents



C.W. Chang, T. Matsui, M. Kondo, J. Non-cryst. solids, 354 (2008) 2365.

- Spin density \Rightarrow photoconductivity
- The smaller Ge-DB density of μ c-Si_{1-x}Ge_x:H is consistent with high photoconductivities

Ge-DB acts as a predominant recombination center

Conclusions

1. Ge incorporation provides an **enhanced infrared light absorption**.
2. Solar cell **performances decreases** with increasing Ge contents.
3. **Ge dangling bonds** are **charged** in large densities due to the presence of the acceptor states in undoped $\mu\text{c-Si}_{1-x}\text{Ge}_x:\text{H}$.
4. The neutral Ge dangling bond acts as a **predominant recombination center**.
5. **Defect states** (dangling bonds) in $\mu\text{c-Si}_{1-x}\text{Ge}_x:\text{H}$ as an acceptor state closed to valence band, which are located at **grain boundaries**.

Ge dangling bond passivation?
Acceptor states compensate?



Thanks for your attention.



Green Energy and Environment Research Laboratories / ITRI

Chia-Wen Chang
changcw@itri.org.tw TEL:03-5915371

Nov. 15, 2010